# The Crystal Structure of $\mathrm{CsI}_{2} \mathbf{B r}$ 

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The structure of $\mathrm{CsI}_{2} \mathrm{Br}$ has been determined by X-ray diffraction. The unit cell is orthorhombic, space group Pmcn, with dimensions $a=6.634(0.008), b=9 \cdot 567(0.012)$, and $c=10.958(0.008) ~ \AA$, where the quantities in parentheses are standard deviations. The cell contains 4 formula units and all atoms lie on mirror planes. The structure was refined by three-dimensional least squares to $R=0.056$. The anion dimensions are $\operatorname{Br}-\mathrm{I}(2)=2.906$ and $\mathrm{I}(2)-\mathrm{I}(3)=2.777 \AA$, and $\mathrm{Br}-\mathrm{I}(2)-\mathrm{I}(3)=178.0^{\circ}$. The compound is isostructural with $\mathrm{CsI}_{3}$.

## Introduction

The structure of $\mathrm{CsI}_{2} \mathrm{Br}$ has been studied for comparison with that of $\mathrm{CsI}_{3}$ (Tasman \& Boswijk, 1955, hereafter referred to as TB; Mooney, 1935) and to verify that the anion is (I-I-Br) $)^{-}$as expected. Cesium triiodide is one of the unsymmetrical triiodides; structural data for related compounds should help to interpret the forces acting to distort the ion and the bonding in the ion. A comprehensive review of the structures of polyhalides has been presented by Wiebenga, Havinga \& Boswijk (1961).

Early optical study of $\mathrm{CsI}_{2} \mathrm{Br}$ and $\mathrm{CsI}_{3}$, as well as other polyhalides, by Wells \& Penfield (1892; hereafter referred to as WP) indicated them to be orthorhombic and isostructural.

## Description and discussion of the structure

X-ray diffraction data show that $\mathrm{CsI}_{2} \mathrm{Br}$ is isostructural with $\mathrm{CsI}_{3}$, the structure of which was determined by Mooney (1935) and refined by TB. Crystal data and atomic parameters are set out in subsequent sections. The structure is illustrated in Fig. 1, and important interatomic distances are given in Table 1.

## Table 1. Interatomic distances in $\mathrm{CsI}_{2} \mathrm{Br}$

All distances are in $\AA$ and are not corrected for the effects of thermal motion. The non-bonded distances are those indicated by dashed lines in Fig.1. They are listed clockwise, starting at the top, around each atom, except that repetitions are omitted.

| $\mathrm{Br}-\mathrm{I}(2)$ | 2.906 | $\mathrm{I}(2)-(\mathrm{I} 3)$ | $2 \cdot 777$ |
| :--- | :--- | :--- | :--- |
| $\mathrm{C} . \ldots \mathrm{I}(3)$ | $4.050(2 \times)$ | $\mathrm{Cs} \ldots \mathrm{Br}$ | 3.581 |
| $\mathrm{Cs} \ldots \mathrm{Br}$ | 3.672 | $\mathrm{I}(2) \ldots \mathrm{I}(3)$ | 4.311 |
| $\mathrm{Cs} \ldots \mathrm{I}(3)$ | 3.993 | $\mathrm{I}(2) \ldots \mathrm{I}(2)$ | $4.407(2 \times)$ |
| $\mathrm{Cs} \ldots \mathrm{Br}$ | $3.748(2 \times)$ | $\mathrm{I}(2) \ldots \mathrm{I}(3)$ | 3.923 |
| $\mathrm{Cs} \ldots \mathrm{I}(3)$ | 3.948 |  |  |
| $\mathrm{Cr} \ldots . \mathrm{I}(3)$ | 3.965 |  |  |

The standard deviation in an interatomic distance amounts to about $0 \cdot 2 \%$ of the distance, when the contributions from axial length errors are included.

The most prominent feature is the triatomic anion $\mathrm{I}_{2} \mathrm{Br}^{-}$. Comparison of the bond lengths in $\mathrm{CsI}_{2} \mathrm{Br}$,

$$
\mathrm{Br}^{2.906 I}(2)^{2.777 \mathbf{I}(3),}
$$

with those in $\mathrm{CsI}_{3}$,

$$
\mathrm{I}(1)^{3.04} \mathrm{I}(2)^{2.83}(3),
$$

shows two features. (1) The $I(2)-I(3)$ bond is shorter in $\mathrm{I}_{2} \mathrm{Br}^{-}$, and thus closer to the bond length $2.68 \AA$ in crystalline $\mathrm{I}_{2}$ (Kitaigorodskii, Khotsyanova \& Struchkov, 1953) or the length $2 \cdot 66 \AA$ in the vapor (Karle, 1955). (2) The $\mathrm{Br}-\mathrm{I}(2)$ bond is longer than would be expected merely from the $I(1)-I(2)$ bond length and the fact that the covalent radius of bromine is about $0 \cdot 20 \AA$ less than that of iodine. These features suggest that the $\mathrm{I}_{2} \mathrm{Br}^{-}$structure is closer to the situation described as $\mathrm{Br}^{-}$plus $\mathrm{I}_{2}$ than is the $\mathrm{I}_{3}^{-}$structure to $\mathrm{I}^{-}$plus $\mathrm{I}_{2}$. This is just what would be expected from the greater electronegativity of bromine relative to iodine. The bromine atom is bonded, nevertheless, to $I(2)$ because their separation is far less than the sum of van der Waals


Fig. 1. The structure of $\mathrm{CsI}_{2} \mathrm{Br}$ projected onto the $b c$ plane. The heavily outlined atoms lie at $x=\frac{1}{4}$, the lightly outlined ones at $x=-\frac{1}{4}$ or $\frac{3}{4}$. $c$-Glide planes are located at $y=\frac{1}{4}$ and $\frac{3}{4}, n$-glide planes at $z=\frac{1}{4}$ and $\frac{3}{4}$. The dashed lines are nonbonded distances; a double dashed line indicates two distances, to atoms above and below the plane at the heavily outlined atoms.
radii, $4 \cdot 1 \AA$. This polarization of $\mathrm{I}_{2} \mathrm{Br}^{-}$inferred from bond lengths receives some support from the fact that the bromine atom occupies a site of higher electrostatic potential than does iodine atom $\mathrm{I}(3)$; the sum of $1 / r$ values for the four nearest cesium ions in each case amount to 1.085 at $\mathrm{Br}, 0.999$ at $\mathrm{I}(3)$. (The corresponding values in $\mathrm{CsI}_{3}$ are 1.046 and 0.971 ). Some feeling for the significance of this difference may be gained by noting that if half an electron, for example, is moved from $\mathrm{I}(3)$ to Br , the electrostatic energy is reduced by $14 \mathrm{kcal} . \mathrm{mole}^{-1}$.
The bond angle $\mathrm{Br}-\mathrm{I}(2)-\mathrm{I}(3)$ is $178.0^{\circ}$, deviating from linearity in the direction which displaces $\mathrm{I}(2)$ toward the center of the unit cell. The corresponding angle is $176.3^{\circ}$ in $\mathrm{CsI}_{3}$.

Table 1 includes all non-bonded distances less than $4 \cdot 5 \AA$. The values must be compared with the corresponding sum of ionic or van der Waals radii, namely, $3.64 \AA$ for $\mathrm{Cs} \cdots \mathrm{Br}, 3 \cdot 84$ for $\mathrm{Cs} \cdots \mathrm{I}, 4 \cdot 10$ for $\mathrm{Br} \cdots \mathrm{I}$, and 4.30 for $\mathrm{I} \cdots \mathrm{I}$. The only distances shorter than the expected van der Waals distances are $\mathrm{Cs} \cdots \mathrm{Br}=$ $3 \cdot 581, \mathrm{Br} \cdots \mathrm{I}(3)=3 \cdot 965$, and $\mathrm{I}(2) \cdots \mathrm{I}(3)=3.923 \AA$. The last of these is the most striking, being almost $0 \cdot 4 \AA$ shorter than the normal van der Waals distance. This, plus the fact that atom $\mathrm{I}(2)$ is not far from collinear with the ion to which $\mathrm{I}(3)$ belongs, suggests a description of the bonding in the group $\mathrm{Br}-\mathrm{I}(2)-$ $\mathrm{I}(3) \cdots \mathrm{I}(2)$ as 4 -center, 6 -electron bonding (Rundle, 1963). Short 'nonbonded' I …I distances are, of course, common. One of the more extreme examples is provided by iodine monochloride in both $\alpha$ (Boswijk, van der Heide, Vos \& Wiebenga, 1956) and $\beta$ forms (Carpenter \& Richards, 1962) where the shortest I … I distances are 3.08 and $3.06 \AA$, respectively.

All interatomic distances quoted previously are uncorrected for the effects of thermal motion. Application of Cruickshank's (1956) analysis of the atomic displacements into rigid-body motions seems to be inappropriate for the anion because the atom (bromine) farthest from the center of mass of the ion does not exhibit the largest amplitudes ( $c f$. Table 2). Application of the Busing \& Levy (1964) analysis is straightforward, but there is no adequate basis here for knowing what kind of correlation is present between the thermal displacements of bonded atoms. For $\mathrm{Br}-\mathrm{I}(2)$, the latter analysis gives $2 \cdot 907 \AA$ as lower bound, 2.942 as upper bound and

## Table 2. Final parameters for $\mathrm{CsI}_{2} \mathrm{Br}$

All atoms lie on the mirror plane at $x=\frac{1}{4}$, and $U_{12}=U_{31}=0$. All values have been multiplied by $10^{4}$. Coordinates are in fractions of cell edges and vibration tensor components in $\AA^{2}$. Beneath each value is the corresponding standard deviation.

| Atom | $y$ | $z$ | $U_{11}$ | $U_{22}$ | $U_{33}$ | $2 U_{23}$ |
| :--- | ---: | ---: | ---: | ---: | ---: | ---: |
| Cs | 4506 | 8357 | 337 | 498 | 390 | -117 |
|  | $3 \cdot 3$ | 2.6 | 46 | 17 | 13 | 26 |
| Br | 3670 | 1628 | 263 | 328 | 286 | -99 |
|  | 4.6 | 3.7 | 68 | 20 | 18 | 32 |
| $\mathrm{I}(2)$ | 5495 | 3749 | 128 | 338 | 325 | 86 |
|  | 2.9 | $2 \cdot 4$ | 46 | 12 | 12 | 22 |
| $\mathrm{I}(3)$ | 7318 | 5721 | 306 | 335 | 421 | -138 |
|  | 3.1 | 2.7 | 47 | 13 | 15 | 23 |

2.924 for uncorrelated motion. Similarly for I(2)-I(3) the values are $2 \cdot 778,2 \cdot 818$, and $2.798 \AA$ respectively. In the absence of more information, the values for uncorrelated motion of the atoms probably represent the best estimates of bond distances here.

## Experimental

## Sample preparation

Crystals of $\mathrm{CsI}_{2} \mathrm{Br}$ were prepared according to the method of WP by adding iodine to cesium bromide dissolved in a mixture of alcohol and water. In view of difficulties experienced previously in attempting to prepare $\mathrm{N}\left(\mathrm{CH}_{3}\right)_{4} \mathrm{I}_{4} \mathrm{Br}$, care was taken to verify that the product was the desired one and not $\mathrm{CsI}_{3}$. Analysis showed I, $51 \cdot 1$ (calc. $54 \cdot 39$ ) $\%$, Br, $17 \cdot 2$ (calc. $17 \cdot 13$ ) $\%$, and preliminary photographs prepared from several medium-sized crystals exhibited distinct differences from photographs of $\mathrm{CsI}_{3}$.

## Unit cell

Preliminary rotation and Weissenberg photographs showed that $\mathrm{CsI}_{2} \mathrm{Br}$ is orthorhombic and isostructural with $\mathrm{CsI}_{3}$, and provided approximate cell dimensions. This confirms the conclusions of WP. The axes are labelled to correspond to those used for $\mathrm{CsI}_{3}$ by TB.

A powder sample, to which silicon powder had been added for calibration purposes, was examined with copper radiation in a Philips diffractometer. On the basis of the approximate cell dimensions, ten powder lines in the range $\theta<17^{\circ}$ could be indexed unambiguously as individual reflections, and two silicon lines appeared in the useful region. The axial lengths were then adjusted by the method of least squares to reproduce as well as possible the measured values of $(\sin \theta)^{2}$. The results are

$$
a=6.634, \quad b=9.567, \quad c=10.958 \AA
$$

with standard deviations respectively $0.008,0.012$, and $0.008 \AA$. Since the indexed lines are at rather small angles, the systematic errors may be as large as the random errors which these figures indicate. Consequently it is estimated that more realistic measures of accuracy are estimated standard deviations double those from least squares, namely $0.016,0.024$, and $0.016 \AA$ in $a, b$, and $c$ respectively. The calculated axial ratios $0 \cdot 6934: 1: 1 \cdot 1454$ agree satisfactorily with values $0 \cdot 6916$ : $1: 1 \cdot 1419$ determined by optical goniometry (WP).
The density of $\mathrm{CSI}_{2} \mathrm{Br}$ may be roughly estimated as $4.4 \mathrm{~g} . \mathrm{cm}^{-3}$ by interpolation between the pycnometric values for $\mathrm{CsI}_{3}$ and CsIBr 2 reported by Bozorth \& Pauling (1925). The X-ray value, on the assumption of 4 formula units per cell, is $4.456 \mathrm{~g} . \mathrm{cm}^{-3}$, thus confirming the assumption.

## Space group

The systematic absences $h 0 l$ for $l$ odd and $h k 0$ for $h+k$ odd imply either space group $P 2_{1} c n$ or $P m c n$ (Pnma relabelled). The crystal morphology (WP) seems
to be centrosymmetric, so Pmen was assumed here, as for $\mathrm{CsI}_{3}$ (TB). The subsequent refinement verifies this choice.

## Structure factor magnitudes

All intensities were measured with a single small well-formed crystal having dimensions $0.15 \times 0.08 \times$ 0.12 mm along $a, b$, and $c$, respectively. It was mounted to rotate about $c$.

Intensities were measured on multiple-film integrated equi-inclination Weissenberg photographs by means of a Nonius microdensitometer. Weak spots were measured by visual comparison of spots on strongly exposed non-integrated photographs with a set of standard spots. Ilford film types Industrial G, B, and C were placed one behind the other in the film holder and exposed simultaneously with zirconium-filtered Mo $K \alpha$ radiation. The layers $h=0,1,2$, and 3 were measured, as well as a special integrated Weissenberg photograph containing a $35^{\circ}$ sample from each layer. This last record enabled the intensity values from the four layers to be placed on the same scale. It also showed that some deterioration of the crystal had occurred during the exposures; the strong spots had become relatively more intense than on earlier films because of reduced extinction. For this reason, as well as because the $x$ coordinates of the structure were known from symmetry grounds, higher $h$ levels were not measured.

The intensities were corrected approximately for absorption by means of a computer program devised for two-dimensional corrections for zero-layer data from crystals with uniform cross-section perpendicular to the rotation axis. It was applied to upper layers by using an increased absorption coefficient to simulate the effect of increased path length. An inadvertent blunder provided an interesting measure of success of this absorption correction. Initially the orientation of the crystal was introduced $90^{\circ}$ away from its correct orientation about the rotation axis so that the longer and shorter crystal dimensions in the cross section were interchanged. When this error was corrected during the refinement, the usual discrepancy index $R$ (for observed reflections only) dropped immediately from $16 \%$ to $10 \%$.

Structure factor magnitudes were then extracted from the corrected intensities in the usual way. Finally the various values from symmetrically equivalent reflections were averaged; in most cases, two or three values contributed to the average. A total of 367 independent reflections were measured.

## The refinement

Since $\mathrm{CsI}_{2} \mathrm{Br}$ is isostructural with $\mathrm{CsI}_{3}$, the atomic coordinates of the latter (TB) were used as a first approximation. All atoms are in special positions 4(c), the mirror planes at $x=\frac{1}{4}$ and $\frac{3}{4}$.

Initial refinement was carried out with $h=0$ data. First the electron density projection was calculated with signs from $\mathrm{CsI}_{3}$ for all but the smaller structure factors. The result appeared generally reasonable and the peak corresponding to $\mathrm{I}(1)$ in $\mathrm{CsI}_{3}$ was significantly lower than the others and so probably bromine. This atom was identified as bromine for a structure factor calculation, and another Fourier synthesis was evaluated with inclusion of all observed structure factors. The refined parameters gave $R=24 \%$ at this stage. Thereafter numerous cycles of isotropic least squares refinement led to nearly complete convergence at $R=8 \cdot 6 \%$. Extra cycles were needed as a result of correcting the absorption corrections (described in the Experimental part) and to verify that the proper atom had been identified as bromine. A parallel refinement with all halogen atoms identified as iodine converged to $R=$ $13 \cdot 2 \%$, a convincingly poorer result.

At this point data from higher layers were included and least-squares refinement was resumed on the Xl computer at Leiden with the block-diagonal leastsquares program of Drs E. Rutten-Keulemans. Convergence was achieved in seven cycles of refinement, including anisotropic temperature factor parameters. A fraction 0.8 of calculated coordinate shifts, 0.6 of calculated temperature parameter shifts, were applied. Subsequently it was apparent that use of full shifts would have accelerated the convergence, so full shifts were applied at the last cycle to obtain the final parameters. The largest coordinate change from the result of the $h=0$ refinement was $0.01 \AA$, and the average change was $0.005 \AA$. The coordinates differ from those of $\mathrm{CsI}_{3}$ by an average of only $0.05 \AA$, but the maximum change, for $y(\mathrm{Cs}) \times b$, amounts to $0.13 \AA$.

The final parameters of the structure are given in Table 2. For all atoms $x=\frac{1}{4}$ and $U_{12}=U_{31}=0$ by symmetry. The standard deviations are those calculated by the least-squares procedure using the diagonal approximation. The labelling of the atoms corresponds to that in Fig. 1 and is like that used for $\mathrm{CsI}_{3}$.

The observed and calculated structure factors are set out in Table 3. A few corrections of then-obvious errors in the observed values have been made since the last cycle of least squares, chiefly rejection of those reflections too weak to be seen on re-examination of the films; the corrected values are marked. Unobserved reflections were not carried through the refinement, on grounds of efficiency. At the end, threshold values were estimated for 150 unobserved reflections by consideration of minimum observed values in neighboring regions of reciprocal space and of the presence of nearby observed reflections to serve as landmarks. For 21 of the 150 unobserved reflections, the calculated structure factor magnitude is larger than the observed, but never as much as $40 \%$ larger. The final value of the conventional discrepancy index, for unweighted observed reflections only, was $R=5 \cdot 6 \%$.

A final difference synthesis, based on the structure factors in Table 3, showed no peaks higher than 1.71

Table 3．Observed and calculated structure factors for $\mathrm{CsI}_{2} \mathrm{Br}$
The values listed are $5 F_{o}$ and $5 F_{c}$ ．The maximum value for a reflection too weak to observe is given a negative value under $F_{o}$ ． A dash in the $F_{o}$ column means that the reflection was obscured by the beam stop．Corrected $F_{o}$ values（see text）are indicated by asterisks．


$$
\mathrm{F}_{0}
$$

$$
\begin{array}{cc}
512 \\
250 & 463 \\
530 & -1272 \\
503 \\
505
\end{array}
$$

$$
\begin{array}{r}
12272 \\
-999 \\
-993
\end{array}
$$

$$
\begin{aligned}
& n \approx \approx \% \\
& n_{n}^{n} \approx 0 \\
& m_{n} \approx 0 \% \\
& n_{n}
\end{aligned}
$$

## 08 00 0   621 246 252 115 <br> $$
\begin{aligned} & 252 \\ & 115 \\ & 1254 \end{aligned}
$$ <br> $\qquad$ <br> $-24$ <br> $$
\begin{aligned} & 254 \\ & 412 \end{aligned}
$$ <br> $$
\begin{array}{r} 73 \\ \hline 08 \end{array}
$$ <br> $$
\begin{array}{cc} 12 & 4 \\ 012 & -9 \\ 12 & -5 \end{array}
$$

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$$
\begin{array}{ll}
9 & 509 \\
6 & -93 \\
6 & -310 \\
6 & -417 \\
6 & -153 \\
7 & -73
\end{array}
$$

$$
\begin{aligned}
& \mathrm{mmmm} \\
& \text { on }
\end{aligned}
$$

$$
\begin{array}{rr}
3 & 3 \\
3 & -1 \\
3 & -1
\end{array}
$$

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and no holes deeper than -2.51 e. $\AA^{-3}$; these extreme values are found near iodine atoms. The estimated standard deviation in the density amounts to about $0 \cdot 66$ e. $\AA^{-3}$ according to the formula of Cruickshank \& Rollett (1953). All atomic centers lie on nearly flat regions.

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# The Crystal Structure of $\mathbf{R h}_{\mathbf{2}} \mathbf{T e}_{\mathbf{3}}$ 

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The compound $\mathrm{Rh}_{3} \mathrm{Te}_{2}$ is orthorhombic, space group Amam, with four molecules in the cell, $a=7 \cdot 694$, $b=12 \cdot 446, c=3.697 \AA$.

The structure is of a new type. There are two kinds of rhodium and two kinds of tellurium atom in the structure. The number and mean lengths of the bonds are: $R h_{I}-5 T e=2.69 \AA, R h_{I}-4 R h=2.84 \AA$, $\mathrm{Rh}_{\mathrm{II}}-5 \mathrm{Te}=2.68 \AA, \mathrm{Rh}_{\mathrm{II}}-5 \mathrm{Rh}=2.94 \AA, \mathrm{Te}_{\mathrm{I}}-8 \mathrm{Rh}=2.73 \AA, \mathrm{Te}_{\mathrm{II}}-7 \mathrm{Rh}=2.66 \AA$.

## Introduction

During a recent visit to the Physics Department of the University of California at San Diego the writer examined X-ray diffraction patterns of numerous preparations in the $\mathrm{Rh}-\mathrm{Te}$ system. It was shown by indexing of the powder pattern that samples obtained by quenching from the melt at the exact composition $\mathrm{Rh}_{3} \mathrm{Te}_{2}$ contained a single phase. Small departures from the ideal stoichiometric ratio resulted in two-phase preparations, the second phase being Rh metal on the Rh-rich side and the high-temperature form (NiAs type) of RhTe (Geller, 1954) on the Te-rich side.

Variations in the lattice spacings of $\mathrm{Rh}_{3} \mathrm{Te}_{2}$ were not observed. It must be concluded that the homogeneity range is very narrow and that the structure is fully ordered.

Upon annealing for days at $450^{\circ} \mathrm{C}_{\mathrm{R}}^{3} \mathrm{Te}_{2}$ is transformed into rhodium metal and a new phase of approximate composition $\mathrm{Rh}: \mathrm{Te} \simeq 1 \cdot 1$. However, this new phase is not identical with the low temperature form of RhTe.

## The structure

The analysis of the powder patterns of $\mathrm{Rh}_{3} \mathrm{Te}_{2}$ gave an orthorhombic base centered translation lattice $A$ with periods.
$a=7.694 \pm 0.004, b=12.446 \pm 0.008, c=3.697 \pm 0.002 \AA$.
The calculated density with four molecules per cell is $\varrho=10.56 \mathrm{g.cm}^{-3}$.
The approximate structure was readily deduced from the powder data with the following results:
Space group: Amam
4 Me in $4(a):(000)\left(\frac{1}{2} 00\right)$
4 Me in $4\left(c_{1}\right): \pm\left(\frac{1}{4} y_{1} 0\right)$ with $y_{1} \simeq 0.53$
4 Me in $4\left(c_{2}\right): \pm\left(y_{2} y_{2}\right)$ with $y_{2} \simeq 0.745$
8 Me in $8(f): \pm\left(x_{3} y_{3} 0\right)\left(\frac{1}{2}+x_{3}, \bar{y}_{3}, 0\right)$
with $x_{3} \simeq 0.06 y_{3} \simeq 0.34$.
However, it was not possible to distinguish between Rh and Te atoms.

Subsequently one single crystal was isolated from one of the quenched samples. The crystal was 2 mm

